



Research Article

# Highly Sensing Properties Sensors Based On Ce-Doped ZnO and SnO<sub>2</sub> Nanoparticles to Ethanol Gas

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## Abstract

A comparative study on the sensing properties of nanoparticles ZnO+ x wt% CeO<sub>2</sub> and SnO<sub>2</sub>+ x wt% CeO<sub>2</sub> (x = 0, 2, 4 and 6) sensors sintered at 400 °C toward ethanol gas have been carried out. The crystal structure and the particle size of the prepared samples were investigated by using XRD, IR and TEM techniques. XRD and IR investigation confirmed that the prepared ZnO and SnO<sub>2</sub> have good crystalline character with average crystallite size of 34.5 and 7.2 nm respectively. The TEM study showed that the particles of the pure ZnO sensor nearly show a hexagonal shape which enhanced by the addition of CeO<sub>2</sub>. While the particles of the SnO<sub>2</sub> sensors displayed fine structures with spherical shape. The electrical conductivity behavior of both oxides samples was nearly similar and the electrical conductivity values of ZnO sensors are higher than that of SnO<sub>2</sub> sensors at the same conditions. The obtained gas sensing results showed that the SnO<sub>2</sub> based sensors have high sensitivity values toward ethanol gas with lower operating temperature than that of ZnO based sensors. On the other hand, ZnO based sensors have slightly rapid response time and short recovery time than that of SnO<sub>2</sub> based sensors.

**Key words:** ZnO nanoparticles; SnO<sub>2</sub> nanoparticles; Ce-doping; ethanol gas sensors

## Introduction

The detection of different types of gases and quantification of their precisely concentration have wide applications in different fields like domestic gas alarms, medical diagnostic equipment, industrial

safety, environmental pollution, military field and food industry [Papadopoulos et al (1996), Wang C et al (2009), Wetchakun et al (2011), Brudzewski et al (2012), Mielle et al (2001) and Fleming (2001)]. With numerous industries that utilize or produce ethanol, it is apparent that reliable methods are needed

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for its detection, monitoring and controlling. Ethanol gas detection with accurately quantification of its concentrations has broad applications in different areas such as monitoring of fermentation, foodstuffs conserving, breath analyzer of drivers to reduce the number of roads accidents caused by excessive alcohol consumption and medical processes [Neri et al (2006), de Lacy Costello et al 2002 and Hieu et al 2010]. Various methods can be used to detect ethanol gas including gas chromatography-mass spectrometry, infrared spectroscopy, semiconductor gas sensors, and others [Pang et al 2001]. Ethanol gas sensors based on semiconductor metal oxide materials offer considerable advantages in comparison to other gas sensing methods [Korotcenkov 2007]. Nano-semiconductor metal oxide gas sensors are inexpensive to produce compared to other sensing technologies, small in size, reliable, fast response, easy to used and can be used in real field. Among the semiconductor metal oxides materials, ZnO and SnO<sub>2</sub> are the two of the most widely materials used in gas sensing applications [Chaisitsak 2011 and Gupta et al 2010]. The advantages of the ZnO and SnO<sub>2</sub> gas sensors are the high sensitivity, stability, novel shapes and low cost. The gas sensing mechanism of the metal oxides is depend on the interaction of tested gas with the pre-adsorbed oxygen species (O<sub>2</sub><sup>-</sup>, O<sup>-</sup> and O<sub>2</sub><sup>-</sup>) on the oxide surface leading to the change in its electrical resistance [Kanan et al 2009 and Wang L et al 2012]. The changes in the electrical resistance can be used as a measure of gas concentration. The gas sensing properties of the semiconductor metal oxide can be improved by preparing the material in the nano-sized range with high surface area which provides more surface active sites on which the gases adsorb and interact. Also, the sensing properties of the metal oxide gas sensor can be enhanced by the addition of some noble metals or other metal oxides [Liewhiran et al 2006, Lee et al 2009, Tianshu et al 1999, Li et al 2012 and Galatsis et al 2003]. Numerous researches have attempted to enhance the gas sensing characteristics of zinc oxide and

tin oxide towards ethanol gas through using some additives such as Pt, Pd, CoO, Al<sub>2</sub>O<sub>3</sub>, CdO and Fe<sub>2</sub>O<sub>3</sub> [Liewhiran et al 2006, Lee et al 2009, Tianshu et al 1999, Li et al 2012 and Galatsis et al 2003]. Rare earth oxides additions are well known to display a high surface basicity, fast oxygen ion mobility and interesting catalytic properties, these are thought to be important for enhancing gas sensing properties of the semiconductor materials [Tsang et al 1998]. The preparation methods, characterization and sensing properties to ethanol gas of ZnO+ x wt% CeO<sub>2</sub> and SnO<sub>2</sub>+ x wt% CeO<sub>2</sub> nanoparticles sensors samples sintered at different temperatures are presented elsewhere [El-Sayed et al 2012 and Hassouna et al 2012]. The current study aims to compare between the samples which exhibited the best sensing characteristics at the same conditions to define the high sensing properties sensors to ethanol gas among the two oxides sensors.

## Experimental

### Synthesis

Zinc oxide nanoparticles were synthesized by chemical precipitation method using zinc acetate dihydrate (99.61%) and diluted ammonium hydroxide as starting materials. The obtained precipitate was calcined in a muffle furnace at the temperature of 400 °C for 4 hours and then left to cool to room temperature. Preparation of tin oxide nanoparticles was carried out also by chemical precipitation method using a solution of tin tetrachloride with concentration of 0.2 mol/L. The precipitate was then filtered and washed thoroughly until free of chloride by testing the filtrate with silver nitrate solution. The obtained precipitate was dried in air followed by calcination at 400 °C for 4 hours then left to cool to room temperature. Appropriate amounts of CeO<sub>2</sub> were added to the prepared ZnO and SnO<sub>2</sub> nanoparticles powders with ratio of 0, 2, 4 and 6 wt % for each. The resulting mixtures were ball milled for 2h to get homogenous powder to be used as functional materials to fabricate sensors

pellets of 12 mm in diameter and 2 mm thick, followed by sintering at 400 °C for 4 hours.

### **Characterization and Measurements**

The prepared samples will be characterized by using X-ray diffraction analysis using X-ray diffractometer (model-Bruker AXS D8 advance) with copper radiation, infrared transmission spectra using Nexus 670 FTIR spectrophotometer (Nicolet, USA) and transmission electron microscope using JEOL JEM-1230 operating at 120 KV attached to CCD camera. a. c. electrical conductivity as well as ethanol gas sensing characteristics of the different sensors were measured in the temperature range from 30 up to 410 °C using LCR meter (Hitester, model Hioki 3532, made in Japan) at frequency of 1 KHz and applied 5 V. The sample chamber was a closed glass cell contains the sensor sample surrounded by electrical cylindrical ceramic furnace with controlled temperature device [Hassona et al 2012]. The desired ethanol gas concentrations were obtained by injection a known volume of ethanol using a micro-syringe through air tight rubber port into the glass chamber. The sensitivity (S) of the sensor toward ethanol gas was measured and calculated as the ratio of the electrical resistance in air atmosphere ( $R_{air}$ ) to that in air containing ethanol gas ( $R_{gas}$ ) by using the following relation [El-Sayed et al 2012 and Hassouna et al 2012].

$$S = R_{air} / R_{gas} \quad (1)$$

## **Results and Discussion**

### **Characterization**

Fig. 1 shows the X-ray diffraction patterns of the prepared zinc oxide and tin oxide. All the diffraction peaks in the patterns can be indexed to hexagonal wurtzite structure of ZnO and tetragonal rutile structure of SnO<sub>2</sub> which is in good agreement with the standard data (card file No. 36-1451 and card file No. 41-1445), respectively. The average crystallite size (D) of the prepared ZnO and

SnO<sub>2</sub> powders are approximately 34.5 nm and 7.2 nm respectively. The X-ray diffraction investigations carried out on the prepared ZnO+ x wt% CeO<sub>2</sub> and SnO<sub>2</sub>+ x wt% CeO<sub>2</sub> sensors and also all sensors sintered at 400 °C confirmed that all the prepared samples are thermally stable and no chemical reaction occurred between ZnO or SnO<sub>2</sub> and the CeO<sub>2</sub> doping materials during the sintering processes. Fig. 2 depicts the averages values of the crystallite sizes of the ZnO + x wt% CeO<sub>2</sub> and SnO<sub>2</sub> + x wt% CeO<sub>2</sub> sensor samples sintered at 400 °C respectively as estimated from the XRD measurement. In general, it was found that the averages of the crystallite sizes of ZnO sensors were higher than that of SnO<sub>2</sub> sensors. Where, the averages of the crystallite sizes of pure and CeO<sub>2</sub>-doped ZnO sensors are ranging from 40.5 to 55 nm, while that of pure and CeO<sub>2</sub>-doped SnO<sub>2</sub> sensor samples are ranging from 11 to 19.2 nm. For both oxides the crystallite size decreased with CeO<sub>2</sub> additions. This indicates that the CeO<sub>2</sub> doping material affects the rate of crystallite growth of ZnO and SnO<sub>2</sub> [Jiang et al. 2010].

The infrared spectra of the prepared ZnO and SnO<sub>2</sub> nanoparticles are shown in Fig. 3. With respect to the prepared ZnO sample, the absorption bands appeared at ~ 3437, 2924, 1516 and 1389 cm<sup>-1</sup> were attributed to O-H stretching mode of the absorbed water, C-H mode of the acetate group, asymmetric and symmetric stretching modes of the carbonate group due to acetate group of starting materials, respectively [Maensiri et al 2006, Kwon et al 2002 and Rezende et al 2009]. The characteristic absorption band of the prepared zinc oxide was observed at ~ 461cm<sup>-1</sup>, which attributed to stretching vibration of Zn-O bond [Maensiri et al 2006, Kwon et al 2002 and Rezende et al 2009]. In the case of SnO<sub>2</sub> sample, Fig. 3(b), the absorption bands appeared at ~3423 and 1632 cm<sup>-1</sup> were assigned to stretching

vibration of -OH group and bending vibration of adsorbed molecular water. The characteristic tin oxide absorption bands were observed at 620 and 540  $\text{cm}^{-1}$  which ascribed to Sn-O-Sn stretching vibration of SnO<sub>2</sub> and terminal oxygen vibration of Sn-OH groups, respectively [Adnan et al 2010 and Mariammal et al 2011]. The observed absorption bands due to adsorbed water and acetate group of starting materials (Fig.3) disappeared with sintering temperature.

The transmission electron microscope images of ZnO + x wt% CeO<sub>2</sub> and SnO<sub>2</sub> + x wt% CeO<sub>2</sub> (x = 0, 2, 4 and 6) sensors sintered at 400 °C are shown in Fig. 4. The particles of the pure ZnO sensor nearly show a hexagonal shape which enhanced by the addition of CeO<sub>2</sub>. While the particles of the SnO<sub>2</sub> sensors displayed fine structures with spherical shape. The averages particle sizes of both oxides samples were estimated and tabulated in Table 1. It can be seen that at the same CeO<sub>2</sub> content the particle size of the ZnO sensors is larger than that of SnO<sub>2</sub> sensors. For both oxides the average particle size were found to decrease with increasing CeO<sub>2</sub> content as indicated in Table 1. These results show that CeO<sub>2</sub> doping material may be acts as an inhibitor which resists the ZnO and SnO<sub>2</sub> grain growth during the sintering temperature [Fen et al. 2007 and Maciel et al. 2003].

### **Electrical Conductivity**

Fig. 5 shows the variation of the electrical conductivity with temperature for ZnO + x wt% CeO<sub>2</sub> and SnO<sub>2</sub> + x wt% CeO<sub>2</sub> sensors sintered at 400 °C. In general, it can be seen that the electrical conductivity increases with temperature and the electrical conductivity behavior of both oxides samples is similar where this behavior is often seen in the case of semiconductor materials. All curves are divided into three different temperature regions denoted as AB, BC and CD which represent the thermal excitation of electron, adsorption of oxygen species and intrinsic semiconductor behavior, respectively [Caglar et al 2009, Sharma et al 2011 and Yu et al

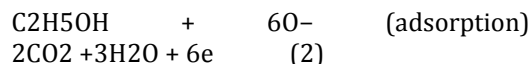
2001]. Also, it can be seen that the electrical conductivity values of ZnO sensors are somewhat higher than that of SnO<sub>2</sub> sensors at the same conditions. This may be due to the difference in the particle and charge carriers concentration and its mobility.

### **Ethanol Gas Sensing Properties**

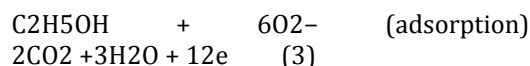
Where the pure and cerium doped sensors of both oxides of ZnO and SnO<sub>2</sub> sintered at 400 °C exhibited the highest sensitivity among all the prepared sensors, for that reason the study focused on these sensors. Fig. 6 illustrates the variation of the sensitivity with temperature towards 100 ppm ethanol gas of ZnO + x wt% CeO<sub>2</sub> and SnO<sub>2</sub> + x wt% CeO<sub>2</sub> sensors sintered at 400 °C. With respect to the pure and CeO<sub>2</sub>-doped ZnO sensors, the sensitivity gradually increases with temperature and attains the maximum 310 °C (operating temperature), and then it decreases with increasing the temperature. While, the sensitivity of the pure and CeO<sub>2</sub>-doped SnO<sub>2</sub> sensors gradually increase with temperature until 300 °C (operating temperature) and then after it decreased. For comparison, it can be seen that the SnO<sub>2</sub> sensors have higher sensitivity values than that of ZnO sensors at the same conditions. Also, the operating temperature of the SnO<sub>2</sub> sensors is lower than that in the case of ZnO sensors by 10 °C.

The mechanism of the ethanol detection on the both oxides sensors is depending on the interaction of ethanol gas with the pre-adsorbed oxygen species on the oxides surface. Where, in atmospheric air the surfaces of ZnO or SnO<sub>2</sub> sensors are covered by oxygen species. At relatively low temperature (< 150 °C) the molecular oxygen species (O<sub>2</sub>) are adsorbed and with increasing the operating temperature the molecular oxygen dissociate to ionic oxygen species (O<sub>2</sub> → 2O → O<sub>2</sub><sup>-</sup>) and its concentration raise gradually until certain temperature by extracting electrons from the metal oxide material which increase the electrical resistance of the materials. The ionic oxygen species O<sup>-</sup> and O<sub>2</sub><sup>-</sup> are the

more reactive species in gas sensing process. When the sensors surface exposed to ethanol gas co-adsorption and mutual interaction between the ethanol gas and the adsorb oxygen species with liberating electrons to the sensors materials which decreases the electrical resistance and the final reaction [El-Sayed et al 2012 and Hassouna et al 2012]:



Or



Where, the changes in the electrical resistance of the sensors in the air to that in gas atmosphere ( $R_{\text{air}}/R_{\text{gas}}$ ) represent the sensitivity of the sensors to the tested gas.

The dependence of the sensitivity on CeO<sub>2</sub> content of ZnO and SnO<sub>2</sub> sensors sintered at 400 °C is shown in Fig. 7. It was found that 4 wt % CeO<sub>2</sub> doping is the more suitable concentration to enhance the sensitivity of ZnO sensor towards ethanol gas. The CeO<sub>2</sub> additions to ZnO sensors up to 4 wt% can promote the ethanol dehydrogenation reaction in the form of catalysts and improve ZnO surface basicity which enhances the sensitivity towards ethanol gas [El-Sayed et al 2012]. On the other hand, the increasing in CeO<sub>2</sub> doping concentration to 6 wt % may be cause a high covering of ZnO surface which reduce the available adsorption sites on ZnO surface and leads to the observed decrease in the sensitivity [Hamedani et al. 2012]. While 2 wt % CeO<sub>2</sub> doping is the more suitable concentration to enhance the sensitivity of SnO<sub>2</sub> sensors towards ethanol gas. The additions of 2 wt % CeO<sub>2</sub> to SnO<sub>2</sub> seem to be benefit for ethanol dehydrogenation on the surface of the sensor sample [Hassouna et al 2012]. Above 2 wt %, CeO<sub>2</sub> content the available adsorption sites on the SnO<sub>2</sub> sensor surface may be reduced which worsen the gas-sensing properties [Pourfayaz et al. 2008].

The variation of the sensitivity with ethanol gas concentration for ZnO + 4 wt % CeO<sub>2</sub> and SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> sensors sintered at 400 °C is shown in Fig. 8. For ZnO + 4 wt % CeO<sub>2</sub> the sensitivity linearly increases with increasing the concentration of ethanol gas up to 400 ppm. While, above 400 ppm, the sensitivity slowly increased. On the other hand, the SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> sensor shows a linear increasing in the sensitivity up to 500 ppm then after it slowly increases with increasing the gas concentration until 2000 ppm.

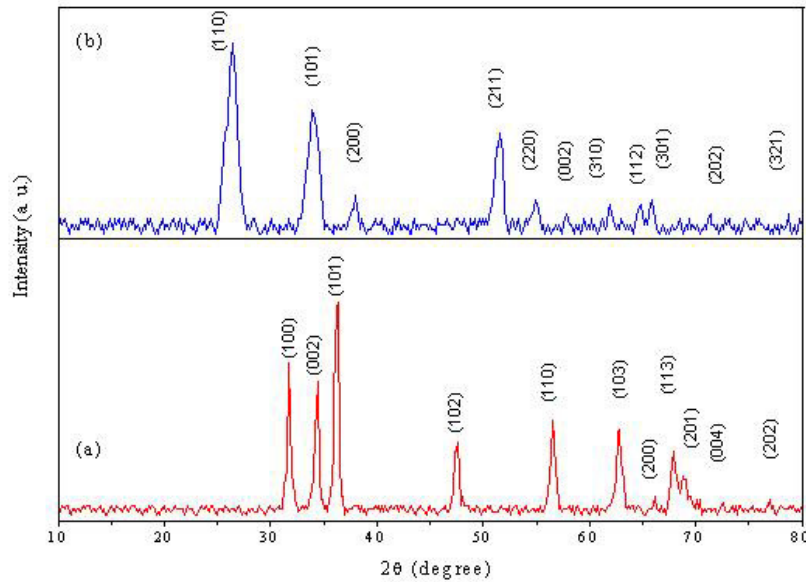
The response time of the sensor is usually defined as the time taken to achieve at least 90 % of the final change in its electrical resistance during exposure to the tested gas. While, the recovery time is generally defined as the time taken by the sensor to get back at least 90 % of its original state after re-exposure to air ambient by maintaining the operating temperature constant [Hassouna et al 2012]. The variation of the sensitivity with time of ZnO + 4 wt % CeO<sub>2</sub> and SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> sensors sintered at 400 °C after exposure to 100 ppm ethanol gas is shown in Fig. 9. It was found that, the response time of ZnO + 4 wt % CeO<sub>2</sub> sensor was 12 second while that of SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> sensor was 20 second. Fig. 10 illustrates the variation of the sensitivity with time of ZnO + 4 wt % CeO<sub>2</sub> and SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> sensors sintered at 400 °C after re-exposure to air atmosphere. It can be seen that, the recovery times of ZnO + 4 wt % CeO<sub>2</sub> and SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> sensors were 10 and 15 second, respectively. The above mentioned results revealed that all the investigated sensors samples are chemically stable i.e there is no chemical reactions occurred between the sensors and the tested gas led to the change in the chemical composition of the sensors during the exposure to ethanol gas.

## Conclusions

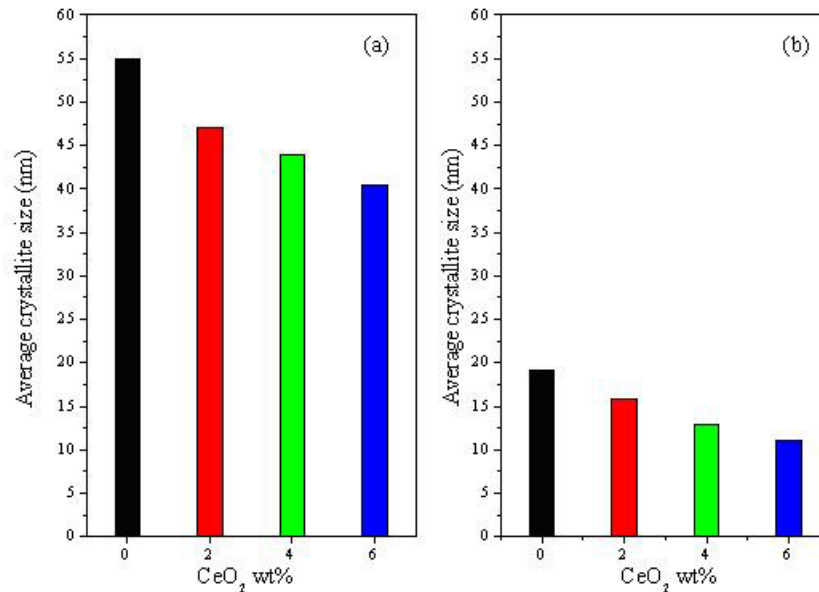
From the obtained data it can be concluded that the best sensors among the prepared sensors of the two oxides are ZnO + 4 wt% CeO<sub>2</sub> and SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> sensors

sintered at 400 °C. The SnO<sub>2</sub> + 2 wt% CeO<sub>2</sub> sensor has high sensitivity towards ethanol gas, rapid response time (20s) and short recovery time (15s) with operating temperature of 300 °C. Also, ZnO + 4 wt%

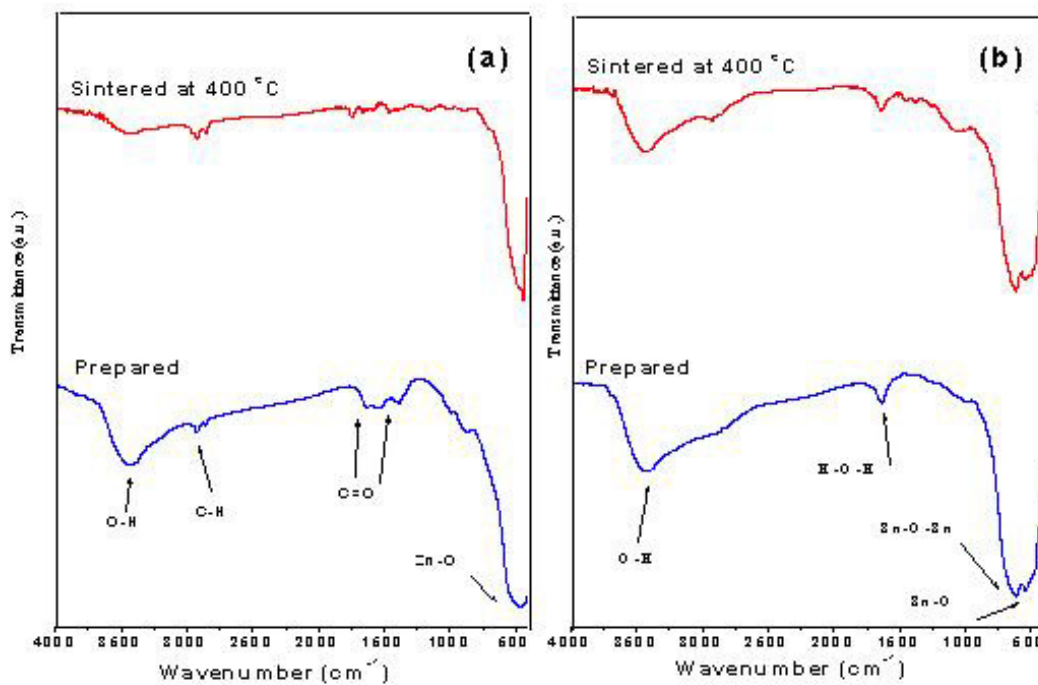
CeO<sub>2</sub> sensor has high sensitivity towards ethanol gas, rapid response time (12s) and short recovery time (10s) with operating temperature of 310 °C.



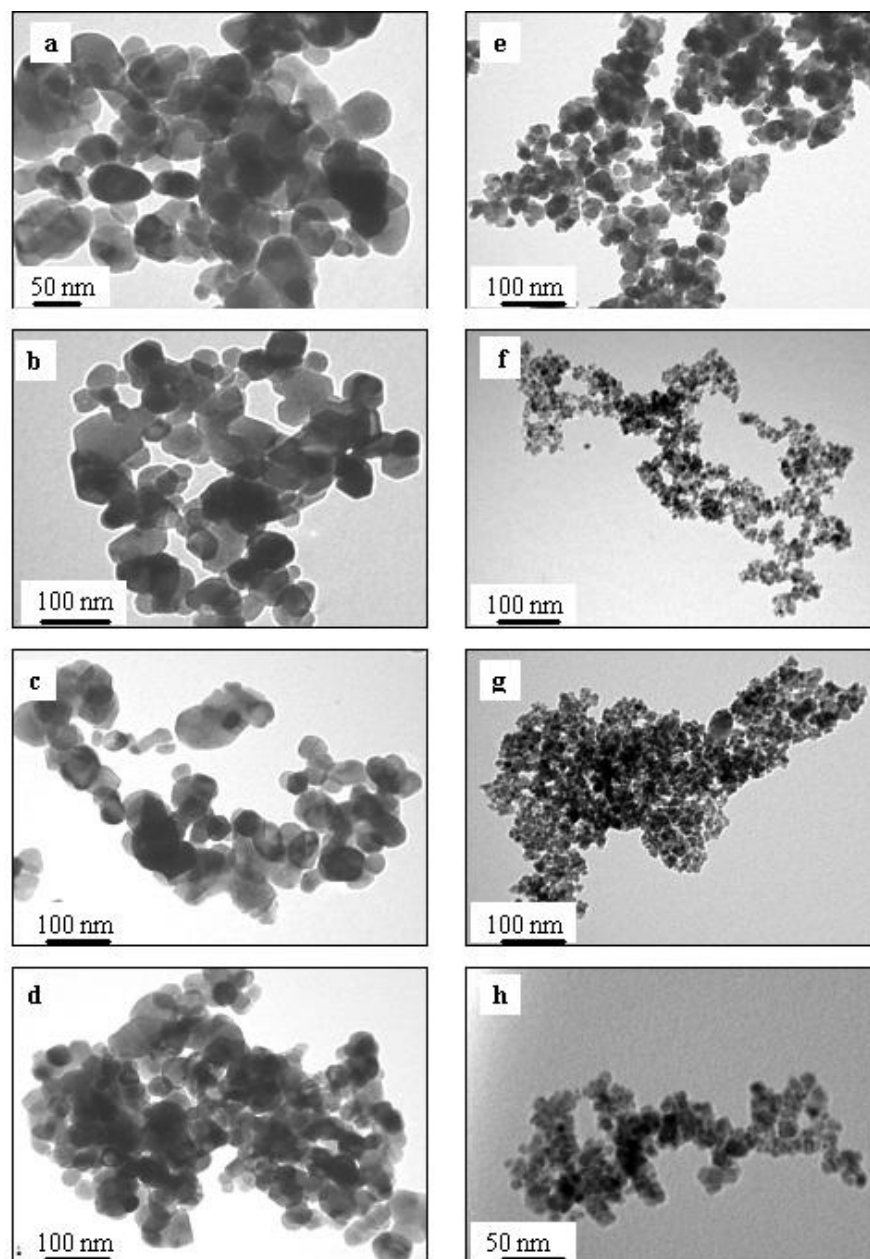
**Figure 1: X-ray diffraction patterns of (a): prepared ZnO and (b): prepared SnO<sub>2</sub> nanoparticles sintered at 400 °C.**



**Figure 2: The averages values of the crystallite size of (a): ZnO + x wt% CeO<sub>2</sub> and (b): SnO<sub>2</sub> + x wt% CeO<sub>2</sub> sensors sintered at 400 °C, where x = 0, 2, 4 and 6.**



**Figure 3: Infrared spectra of (a): prepared and sintered ZnO nanoparticles samples and (b): prepared and sintered SnO<sub>2</sub> nanoparticles samples.**



**Figure 4: TEM images of ZnO + x wt% CeO<sub>2</sub> sensors where (a) x = 0, (b) x = 2, (c) x = 4 and (d) x = 6, and SnO<sub>2</sub> + x wt% CeO<sub>2</sub> where (e) x = 0, (f) x = 2, (g) x = 4 and (h) x = 6 sensors sintered at 400 °C.**



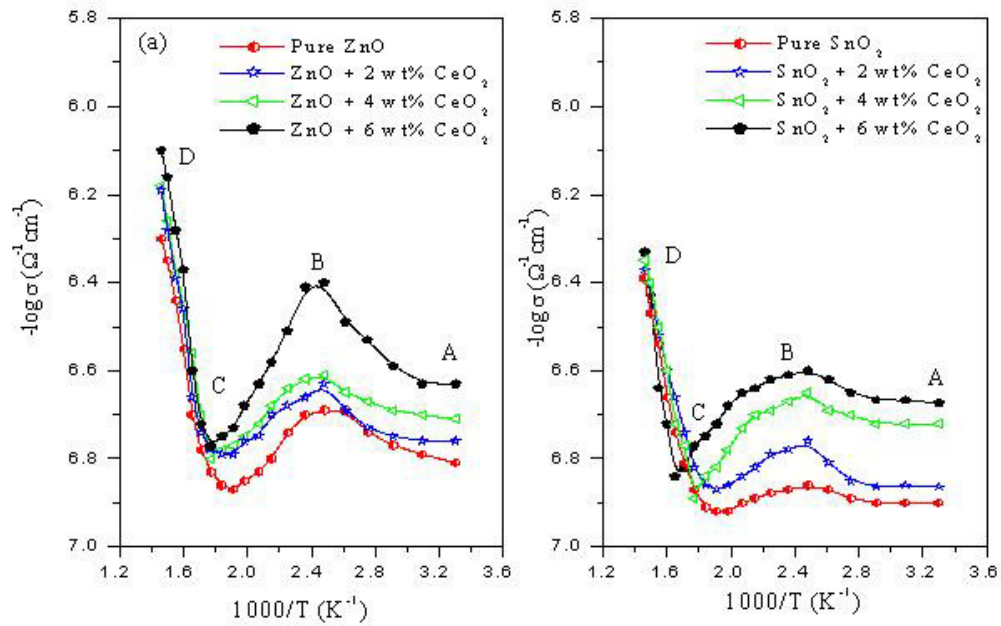


Figure 5: The variation of the electrical conductivity with temperature of (a):  $\text{ZnO} + x \text{ wt}\% \text{CeO}_2$  and (b):  $\text{SnO}_2 + x \text{ wt}\% \text{CeO}_2$  sensors sintered at  $400^\circ\text{C}$ .

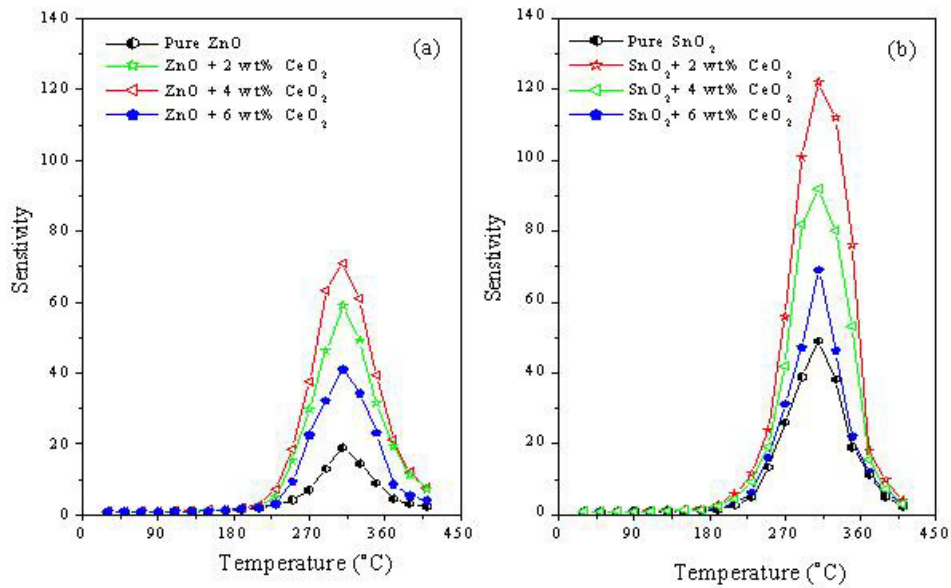
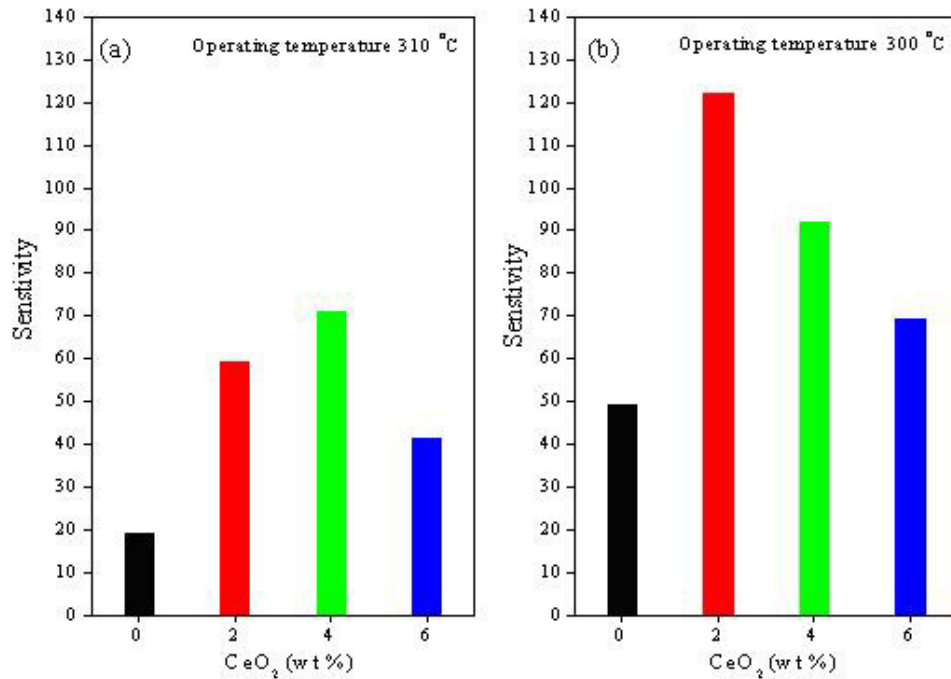
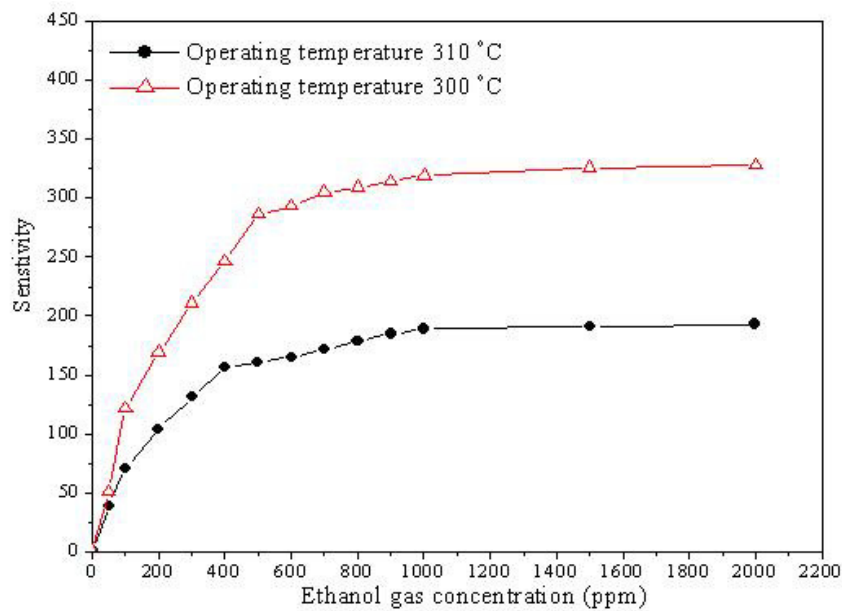


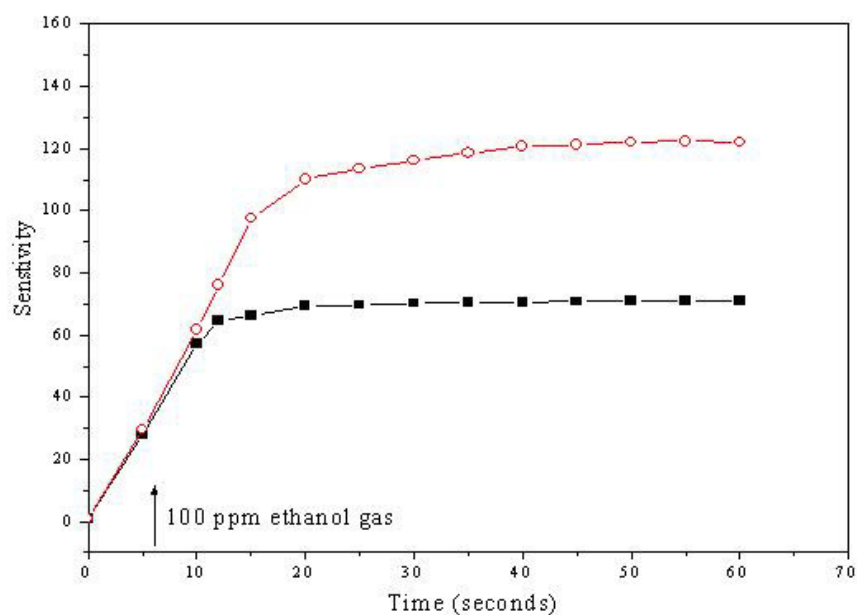
Figure 6: The variation of the sensitivity with temperature of (a):  $\text{ZnO} + x \text{ wt}\% \text{CeO}_2$  and (b):  $\text{SnO}_2 + x \text{ wt}\% \text{CeO}_2$  sensors sintered at  $400^\circ\text{C}$ , 100 ppm ethanol gas.



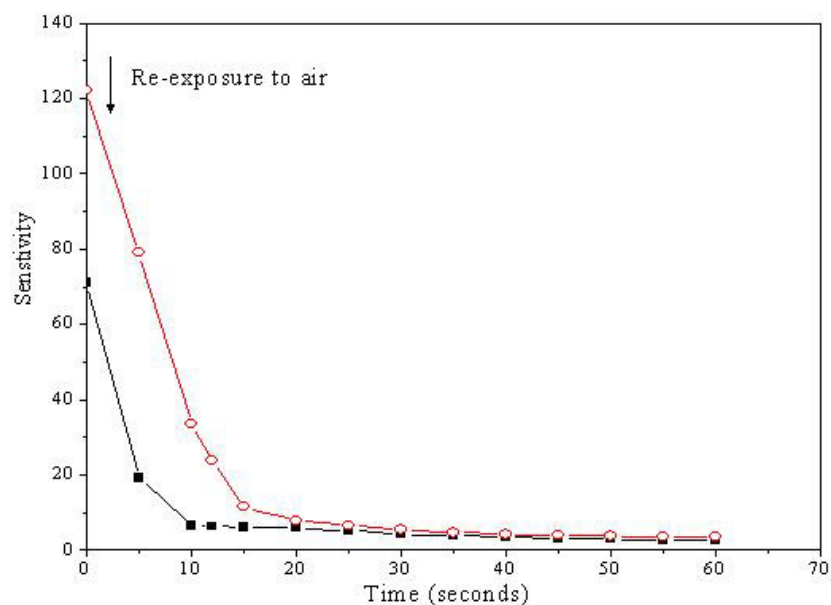
**Figure 7: The dependence of the sensitivity on CeO<sub>2</sub> content of (a): ZnO + x wt% CeO<sub>2</sub> and (b): SnO<sub>2</sub> + x wt% CeO<sub>2</sub> sensors sintered at 400 °C, where x = 0, 2, 4 and 6.**



**Figure 8: The variation of the sensitivity with ethanol gas concentration for (●): ZnO + 4 wt % CeO<sub>2</sub> and (Δ): SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> sensors sintered at 400 °C.**



**Figure 9:** The variation of the sensitivity with time of (■) ZnO + 4 wt % CeO<sub>2</sub> (operating temperature 310 °C) and (○) 32SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> (operating temperature 300 °C) sensors sintered at 400 °C during exposure to ethanol gas.



**Figure 10:** The variation of the sensitivity with time of (■) ZnO + 4 wt % CeO<sub>2</sub> (operating temperature 310 °C) and (○) SnO<sub>2</sub> + 2 wt % CeO<sub>2</sub> (operating temperature 300 °C) sensors sintered at 400 °C after re-exposure to air atmosphere.

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